# Processing scanning Laue microdiffraction patterns with machine learning algorithms: A case-study with fatigued polycrystalline copper

## Introduction

Laue diffraction, that may occur when a polychromatic X-ray beam illuminated a crystal, was first discovered in 1912, and has revealed both the electromagnetic nature of X-ray and the periodic ordering of atoms in crystal (Eckert, 2012). Thanks to the polychromaticity of the employed X-ray, multiple diffraction peaks can be recorded in a single exposure without any rotation that might lead to the ambiguity of the illuminated volume (Chung & Ice, 1999). With the development of polychromatic beam focusing optics, notably Kirkpatrick–Baez mirrors, micron-sized high-brilliance polychromatic X-ray beam can be produced at synchrotron radiation sources and directed to probe inside materials with submicrometric spatial resolution, i.e. Laue microdiffraction. In analogy to EBSD (electron backscatter diffraction) technique, Laue microdiffraction technique maps the lattice orientation and distortion from the one-by-one analysis of the diffraction pattern emanating from each scanned spot (Spolenak *et al.*, 2003, Tamura *et al.*, 2003, Zhou *et al.*, 2018). The two techniques are comparable (Plancher *et al.*, 2016) and complementary to each other (Örs *et al.*, 2018). It is generally accepted that EBSD has an edge on finer spatial resolution of nanoscale, whilst Laue microdiffraction can have a much better accuracy on the lattice orientation and distortion with an attainable order of 10-9 (Zhang, Bieler*, et al.*, 2018).

A salient feature of Laue microdiffraction is its sensitivity to the local misorientation in the illuminated volume (Barabash *et al.*, 2001, Barabash *et al.*, 2003), more specifically, the fragmentation of Laue spot may indicate the presence of geometrically necessary boundaries and the elongation of Laue spot the presence of geometrically necessary dislocations. Although a critical aspect of the spot shape analysis lies on the assumption that the dislocations were dominantly edge-type in the illuminated volume, a recent study with focused ion beam transmission electron microscopy confirmed that this analysis stood still if the dislocations had predominately screw-type (Zhang, Balachandran*, et al.*, 2018). With the aid of a wire profiler (typically Pt), the shape of spot can be spatially resolved to yield subsurface, 3D mapping of lattice orientation and distortion non-destructively (Yang *et al.*, 2004, Barabash *et al.*, 2009, Das *et al.*, 2018), namely the differential-aperture X-ray microscopy (DAXM) technique (Larson *et al.*, 2002).

Despite the wealth of information behind Laue microdiffraction pattern, the interpretation of Laue microdiffraction pattern is not straightforward as the wavelength or indexation pertaining to each diffraction peak is unknown. Standard treatment involves modulating the orientation and calibration parameters to minimize the discrepancy between the simulated and experimental diffraction pattern, and has been implemented in software such as *XMAS* (Tamura, 2014) and *LaueTools* (<https://gitlab.esrf.fr/micha/lauetools>). The standard treatment is in essence a trial-and-error process, and any additional information concerning the scanned microstructure would be favored, for example, Örs *et al.* (2018) used the orientation obtained by EBSD to overcome the difficulty in indexing the Laue microdiffraction patterns. The interpretation process can also be facilitated and improved with the emerging image processing techniques: Petit *et al.* (2015) introduced the digital image correlation (DIC) technique to have a better measurement of relative lattice distortion with reference to an assumed stress-free position, i.e. Laue-DIC method, Zhang *et al.* (2015) later extended the framework of Laue-DIC to get rid of the dependence on the stress-free reference; Zhang *et al.* (2019) used DIC to correct the misalignment of the investigated volume in the experiments of DXAM; Song *et al.* (2019).

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